

## Identification and Quantification of Boron Dopant Sites in Antiferromagnetic Cr<sub>2</sub>O<sub>3</sub> Films by Electron Energy Loss Spectroscopy

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Cr<sub>2</sub>O<sub>3</sub> is antiferromagnetic in bulk but ferromagnetic on the (0001) surface. Bulk Cr<sub>2</sub>O<sub>3</sub> has two degenerate antiferromagnetic states that can be switched by an electric field, which also switches the (0001) surface magnetism [1], creating the potential for voltage-controlled magnetic data storage and spintronic data processing. However, the magnetic ordering (Néel) temperature of pure Cr<sub>2</sub>O<sub>3</sub> is too low for microelectronic applications. Boron substituting for oxygen in Cr<sub>2</sub>O<sub>3</sub> to form BCr<sub>4</sub> tetrahedra increases the exchange energy of neighboring Cr atoms from the favorable hybridization geometry [2], increasing the Néel temperature.

We have combined electron energy loss spectroscopy (EELS) fine structure and density functional theory (DFT) calculations to determine the structure of B dopants in Cr<sub>2</sub>O<sub>3</sub> thin films. EELS is highly sensitive to the nature of chemical bonds and to an atom's coordination environment. Especially for doped oxides when the dopant is surrounded by strong backscattering O atoms, EELS is dominated by scattering events within the local neighbours that can be explained by a molecular approach [3]. We find B in BCr<sub>4</sub> tetrahedra, but also in distorted BO<sub>3</sub> triangles. The BO<sub>3</sub> triangles can be identified by manipulating the electron momentum transfer ( $q$ ) in EELS, which changes the differential cross sections for the 1s- $\pi^*$  and 1s- $\sigma^*$  transitions. BCr<sub>4</sub> tetrahedra can be identified and also quantified by linear combination of simulated B K edges to reproduce the experimental  $\pi / \sigma$  ratios.

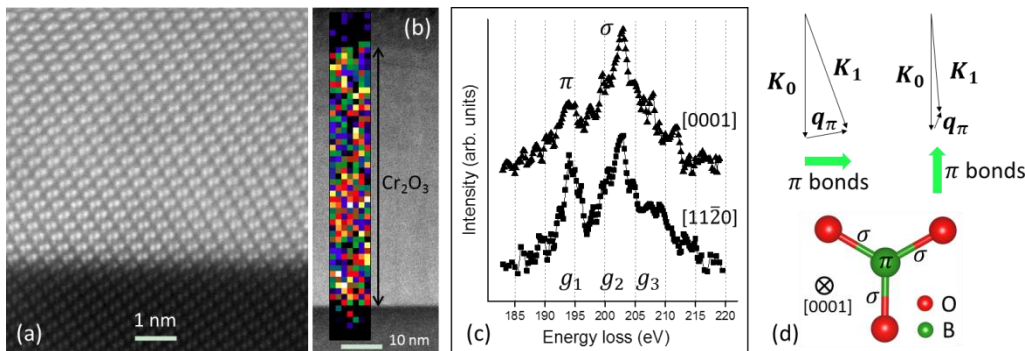
Figure 1(a) is a HAADF STEM image of a 2.5% B doped Cr<sub>2</sub>O<sub>3</sub> film on Al<sub>2</sub>O<sub>3</sub> substrate. Both the B and O atoms are invisible due to relatively weak high-angle scattering cross section. Figure 1(b) is a map of the B K edge intensity, showing a uniform distribution of B in the Cr<sub>2</sub>O<sub>3</sub> film. Figure 1(c) is the B K edges acquired from plan view (top) and cross section (bottom) zone axes. The 194 eV peak corresponds to the 1s- $\pi^*$  transition and the 203 eV peak corresponds to the 1s- $\sigma^*$  transition [4]. The typical geometry of the B sp<sup>2</sup> hybridization forms BO<sub>3</sub> triangles with  $\sigma$  bonds parallel to the BO plane and  $\pi$  bonds normal to the BO plane [3]. The B K edges show clear orientation dependence because of the significant directionality of unoccupied 2p states of the BO<sub>3</sub> sp<sup>2</sup> structure. As illustrated in Figure 1(d), with the large EELS collection angles used in this experiment, the  $\pi$  intensity is proportional to the projection of the momentum transfer  $q$  onto the corresponding orbital orientations [4]. This indicates the BO<sub>3</sub> plane is normal to the [0001] direction.

Figure 2(a) is the possible B dopant structures, relaxed by DFT. All the structures shown have relatively low formation energies ( $\pm 2$  eV) calculated by Zhang-Northrup formalism [5] under film growth conditions, so they are thermodynamically accessible. The simulated BO<sub>3</sub> structures are either planar or slightly distorted, but all of them are normal to the [0001] directions, which matches with the EELS results in Figure 1(d). The existence of BCr<sub>4</sub> can be confirmed by the distinct  $\pi / \sigma$  ratios from multiple-scattering EELS simulation in Figure 2(b), especially on the plan view zone. The bounds of the

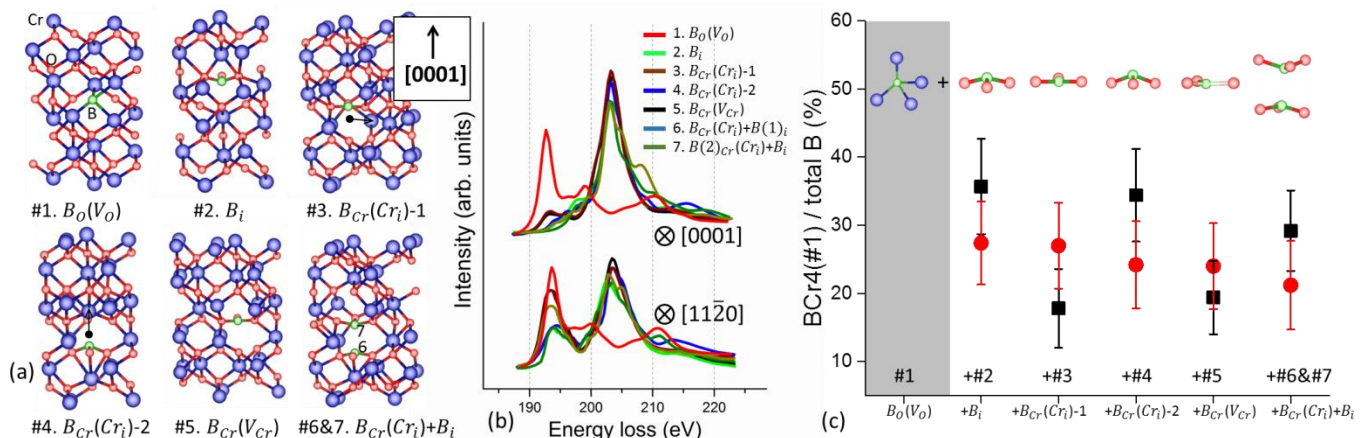
functional  $\text{BCr}_4$  out of the total B can be probed by linear combination of simulated  $\text{BCr}_4$  K edges with each  $\text{BO}_3$  K edge to reproduce experimental  $\pi / \sigma$  ratios, which is shown in Figure 2(c). Only 12%-43% B dopants form  $\text{BCr}_4$  tetrahedra and increases Néel temperature. Further calculation shows that the other  $\text{BO}_3$  structures can frustrate the bulk antiferromagnetic order and suppress the interface exchange bias. To make B doping more efficient, posting annealing under low oxygen partial pressure will favor the formation of  $\text{BCr}_4$ , and co-doping is also a promising approach to stabilize  $\text{BCr}_4$  [5].

References:

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**Figure 1.** (a) HAADF STEM image of a B doped  $\text{Cr}_2\text{O}_3$  film on  $\text{Al}_2\text{O}_3$  substrate. (b) EEL spectrum guide image showing the whole 80 nm thick  $\text{Cr}_2\text{O}_3$  film with B K edge intensity map superimposed. (c) B K edges from plan view and cross section zone axis. (d) Schematic illustration of EELS momentum transfer and the suggested orientation of  $\text{BO}_3$  triangles.



**Figure 2.** (a, and b) Calculated B dopant structures and corresponding EELS from multiple scattering simulations. (c) Quantification of the functional  $\text{BCr}_4$  tetrahedra (#1.  $B_O(V_O)$ ) / total B by linear combination of simulated  $\text{BCr}_4$  K edges with each  $\text{BO}_3$  triangles K edges to reproduce the experiment  $\pi / \sigma$  ratio. Black symbols are from the cross section data, and red symbols are from plan view data.