Characterization of Nano-scale Instabilities in Titanium Alloys Using Aberration-Corrected Scanning Transmission Electron Microscope

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Due to the refined nature of microstructures that can be effectively manipulated by the application of various thermal/mechanical processes, metastable beta titanium alloys have attracted considerable attention in recent days. Usually, such refinement involves the precipitation of the intragranular *hcp* structure alpha phase. In authors recent studies, it has been shown that the size, morphology and number density of these alpha precipitates in Ti-5Al-5Mo-5V-3Cr (Ti-5553, wt.%) can be significantly influenced by the nano-scale structural and compositional instability present in parent *bcc* structure beta phase, more specifically in this alloy, the metastable *hexagonal* structure isothermal omega phase [1-3]. In these latter studies, it was found that either the compositional and/or stress field associated with the isothermal omega phase may contribute to an increased driving force for alpha nucleation [3]. Recent technological improvement in TEM's, spectroscopy detectors and cameras, specifically probe aberration corrected STEM instruments, have enabled atomic resolution z-contrast high angle annular dark field-scanning transmission electron microscopy (HAADF-STEM) imaging capable of characterizing atomic column configurations with a sub-angstrom probe [4] and provides novel insights of new nano-scale instabilities in titanium alloys.

In the first part of this current work, the crystallography and the structure of a previous unidentified ordered orthorhombic metastable phase, termed as O" phase, in the beta matrix of Ti-5553 were studied using probe corrected scanning transmission electron microscope (FEI TitanTM 80-300) [5]. For the first time, using Z-contrast HAADF-STEM imaging, a novel ordering mechanism in beta titanium alloy, that one of every three {110} beta planes exhibits a relative lower intensity than the other two as shown in Fig 1(a), is clearly observed and validated. The lattice parameters of O" phase are a=0.328 nm, b=0.464nm and c=1.393nm, with α = β = γ =90° and O" phase obeys the following OR with the parent β matrix: $(001)_{O^{-}}/(011)_{\beta}$ and $[100]_{O^{-}}/[100]_{\beta}$, shown in Fig 1(b).

The second part of the work to be presented focused on the characterization of a newly characterized nano-scale sturcutual instability in Zr added Ti-Nb beta type titanium alloy [6]. This new phase, termed as 0' phase, was investigated using probe corrected scanning transmission electron microscope (FEI TitanTM 80-300). A high magnification HAADF image is shown in Fig. 2(a), in which it is clear that in the lower left part of the image, bounded by the dashed red box, the *bcc* symmetry has been broken by small displacements of every other atomic column. These displacements result in a structure very similar to that shown in the schematic diagram in Fig. 2(b), which depicts the result of the application of the $\{110\} < 1\overline{10} > \text{shuffle}$, i.e., a disordered orthorhombic structure.

References:

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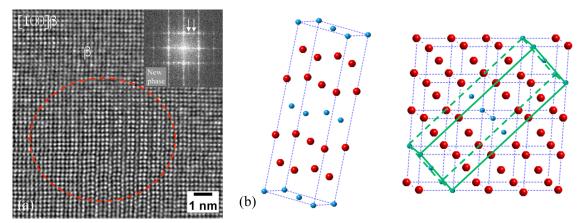


Figure 1. (a) Z-contrast HAAD-STEM image showing the nano-scale ordered orthorhombic structure O" phase; (b) schematic figure showing the structure of O" phase and its orientation relationship with beta phase matrix [5].

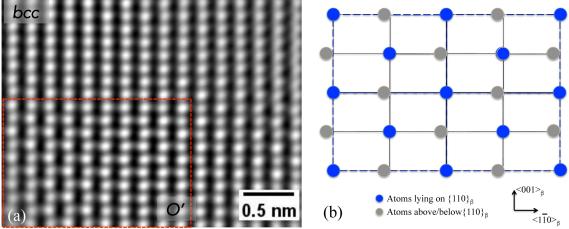


Figure 2. (a) Z-contrast HAAD-STEM image showing the nano-scale disordered orthorhombic structure O' phase; (b) schematic figure showing the structure of O' phase [6].