

Atom Probe Tomography and analytical Scanning Transmission Electron Microscopy of Rapid Solidification Microstructures in Al-Cu Alloy Thin Films

Jörg M.K. Wiezorek¹, Kai W. Zweiacker^{1,2}, Can Liu¹, Isabelle Martin³, Ty. J. Prosa³, David J. Larson³

¹ Department of Mechanical Engineering and Materials Science, Swanson School of Engineering, University of Pittsburgh, Pittsburgh, PA, USA.

² Now: EMPA, Swiss Federal Laboratories for Materials Science and Technology, Dübendorf, Switzerland.

³ CAMECA Instruments Inc., Madison, WI, USA.

Laser processing of alloys results in formation of rapid solidification (RS) microstructures that can exhibit refined length scale, extreme solute segregation, formation of meta-stable phases, and unusual textures [1]. Recent nano-scale spatio-temporal resolution *in situ* dynamical transmission electron microscopy (DTEM) studies of pulsed laser (PL) induced RS in multi-component alloys of hypo-eutectic Al-Cu thin film specimens have identified characteristic morphological modulations of the multi-phase microstructures with solidification interface velocity [1, 2]. Elucidating constitutional effects during RS crystal growth in multi-component systems is of fundamental importance in understanding laser-assisted materials processing [1-5]. However, due to the nano-scale of the resulting microstructural features, acquiring quantitative analytical data with the appropriate spatial-resolution is challenging with scanning and transmission electron microscopes (SEM and TEM). A promising alternative analytical technique is atom probe tomography (APT). APT offers sub-nanometer spatial resolution, and provides statistically significant representative data sets for elemental composition mapping, but is limited to relatively small analytical volumes. The current work presents complementary aberration-corrected scanning TEM (STEM) based energy dispersive X-ray spectroscopy (EDXS) mapping to study constitutional effects in RS microstructure evolution of hypo-eutectic Al-Cu alloys.

Al-Cu thin film alloy RS microstructures were generated by PL irradiation of the electron transparent regions of silicon nitride membrane supported windows [1, 2, 6]. APT specimens were prepared using focused ion beam methods [7] from regions at the perimeter of the PL induced melt-pool (~25 μ m by ~230 μ m, dotted line Fig. 1a). High RS interface velocities, up to ~2.0 m/s, generate four distinct morphological zones: A heat affected zone (HAZ, Zone 1), a narrow transition region (Zone 2) with elongated α -Al grains, the RS crystal growth region with columnar morphology cellular two-phase growth (Zone 3a), and the banded-morphology growth region (Zone 3b) (Fig. 1b) [2, 6]. Under equilibrium conditions hypo-eutectic Al-Cu contains the face-centered cubic α -Al matrix (up to 2.6at%Cu) and tetragonal θ -Al₂Cu (~33at%Cu) phase. The DF-STEM image inset in Figure 1b of the melt-pool edge shows the columnar growth zone (Zone 3a) with discontinuous fine-scale copper enriched metastable Al₂Cu (θ') phase, the transition region with elongated α -Al grains (Zone 2) and the Cu-enriched continuous network of θ -Al₂Cu phase in Zone 1 [6]. Cu%-iso-surfaces and proxigrams for APT specimen M08 (Zone 1) and M05 (Zone 2 / Zone 3a) reveal clearly two distinct phases, corresponding to supersaturated α -Al (~4at% Cu) and Al₂Cu related phase (~33at%Cu for Zone 1 and ~42at%Cu for Zone 2 / Zone 3a), respectively (Figs. 2a, 2b). APT and STEM EDXS analyses showed nearly identical Cu% for the α -Al phase in Zones 1, 2 and 3a (Fig. 2). STEM EDXS measurements tended to underestimate the Cu-content of the Al₂Cu related phase, since this minority constituent is typically embedded in the majority matrix phase of α -Al. However, STEM EDXS analyses of the continuous layers of Cu-enriched phase in Zone 2 (Figs. 2c, 2d) showed enhanced Cu% with on average

39.7at%Cu. In the Zone 2 a rapid acceleration of the previously stagnant solid-liquid interface to ~ 0.8 m/s occurs after a significant incubation time following PL melting [6]. The composition variations in the α -Al phase and the Cu-enriched Al_2Cu related phase in Zones 1, 2 and 3a, inclusive of the local increase beyond 33at%Cu expected for stoichiometric Al_2Cu , represent clear deviations from equilibrium crystal growth. The Al-Cu alloy composition analyses by APT and STEM will be discussed and correlated with the solid-liquid interface velocity evolution during the PL induced RS [8].

References:

- [1] J.E. Kline and J.P. Leonard, Applied Physics Letters **86** (2005), p. 201902.
- [2] J.T. McKeown *et al*, Acta Materialia **65** (2014), p. 56.
- [3] S.C. Gill *et al*, Acta Metallurgica Materiala **40**(1992), p. 2895.
- [4] S.C. Gill and W. Kurz, Acta Metallurgica Materiala **41**(1993), p.3 563.
- [5] W. Kurz and P. Gilgen, Materials Science and Engineering A **178**(1994), p. 171.
- [6] J.T. McKeown *et al*, JOM **68** (2016), p. 985.
- [7] D. J. Larson *et al.*, “Local Electrode Atom Probe Tomography”, (Springer, New York) (2013).
- [8] The authors acknowledge funding from the National Science Foundation, Grants NSF-1105757 and NSF-1607922.

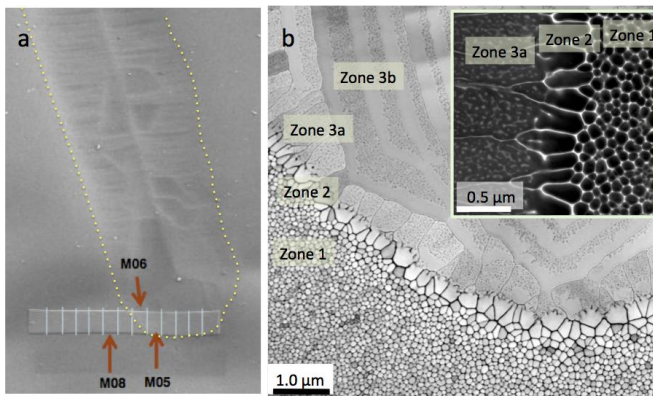


Figure 1. a) SEM secondary electron image of RS region in Al-10Cu alloy thin film (inside dotted outline) and location for APT specimen extractions, e.g. M08 (Fig. 2a) and M05 (Fig. 2b), b) STEM BF and as inset enlarged detail STEM HAADF of the RS microstructure zones: Zone 1, Zone 2, Zone 3a and Zone 3b.

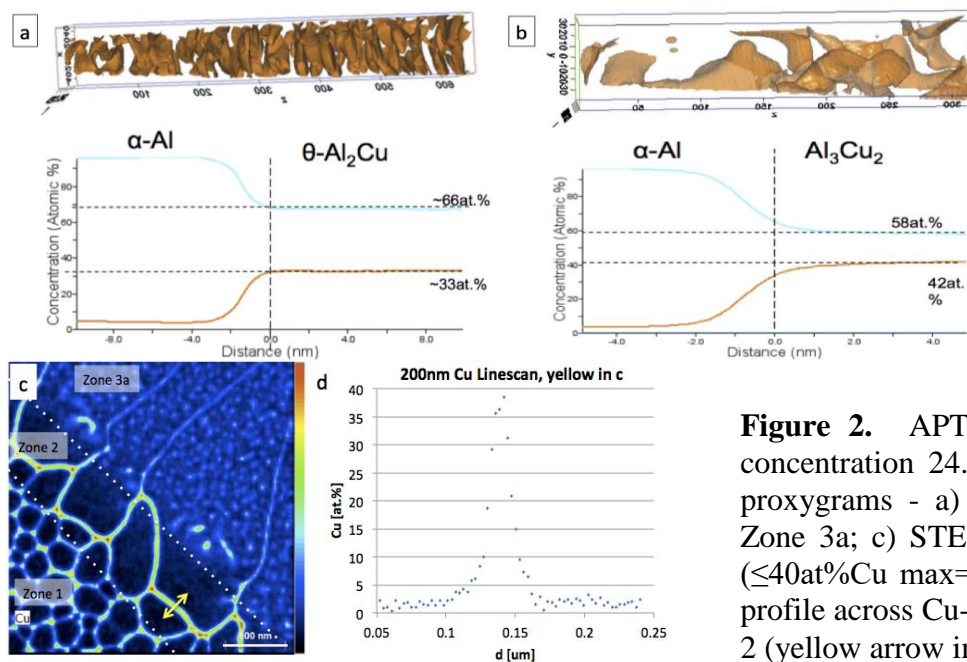


Figure 2. APT Cu ion map for iso-concentration 24.6at%Cu and associated proxigrams - a) Zone 1, b) Zone 2 to Zone 3a; c) STEM EDXS Cu heat map ($\leq 40\text{at}\% \text{Cu}$ max=red); d) Cu EDXS line profile across Cu-enriched region in Zone 2 (yellow arrow in c).