

Fast Atomic-Scale Chemical Imaging by STEM EDS for Study of Crystalline Materials and Dynamic Phase Transformations

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Chemical imaging at the atomic-scale provides a useful tool to investigate material structures and has been recently demonstrated in aberration corrected scanning transmission electron microscopy (STEM). By scanning an angstrom-sized electron probe across a crystal sample aligned along a low-axis orientation and collecting either electron energy-loss spectra (EELS) or energy-dispersive x-ray spectra (EDS), atomic-scale chemical imaging has been achieved. Atomic-scale STEM EDS chemical imaging offers a benefit of easy interpretation with a one-to-one correspondence between image and structure, however has a severe shortcoming due to the poor efficiency of x-ray generation and collection. To compensate the poor efficiency, spectral imaging STEM EDS datasets are often acquired as a series of frames, where the same region is scanned multiple times (typically over few 100 times) and spatially drift-corrected to build a dataset [1, 2]. A typical data collection requires more than a few 100 seconds, subjecting the samples to high irradiation of electrons. This limits the materials that can be studied by the technique to only those that are electron radiation-resistant, and prevents the study of phase transformations and dynamics, which requires the atomic-scale chemical structural information resolved on a much shorter time scale.

In this study, we report a novel approach that allows realization of rapid atomic-scale chemical imaging by STEM EDS. Using LaAlO₃ (LAO) as a model crystal, we find averaged atomic-scale chemical maps for La and Al atoms can be extracted from an area of ~ few 10 nm² acquired with the acquisition time of ~2 seconds or less.

Figure 1 illustrates the method. The x-ray counts, represented by red dots, are distributed on a tile pattern which represents a crystal lattice that gives rise to the x-rays when electron probe is scanned over the region (Fig.1a). Under the atomic-scale STEM EDS imaging conditions, the x-ray counts, which might be seemingly random and sparsely scattered under a short acquisition time (i.e., less than few seconds), are not really random but statistically localized to the atomic columns. The statistical distribution or chemical map (Fig.1c) can be obtained by translating the x-rays distributed over the large region to the equivalent pixel positions within a single unit cell via lattice-vector translation, as illustrated in Figs.1a and 1b. By concentrating x-ray signals into the unit cell, an acquisition time savings of over 100 times can be achieved, and an averaged chemical map can be obtained with the acquisition time of less than 1 sec.

Figure 2a shows experimental x-ray maps from a LAO crystal in [100] which is acquired with a total acquisition time of about 1.8 sec from a region of 4.46 nm×4.46 nm. The maps are overlaid on the HAADF image acquired simultaneously. Using the lattice-vector translation method as illustrated in Fig.1, chemical maps for Sr and Ti can be obtained (Figs.2b and 2c), respectively, and the RGB map obtained (Fig.2d, Sr in red, Ti in green) clearly shows the structure of the LAO (Fig.2e). We have since applied the method to study several different crystals such as SrTiO₃, and GaN, and have verified the

technique using statistically based phenomenological modeling [3]. The fast atomic-scale chemical mapping has also been applied to study the electron-beam induced phase transformation in a layered lithium transition-metal oxide, $\text{Li}[\text{Li}_{0.2}\text{Ni}_{0.2}\text{Mn}_{0.6}]\text{O}_2$ (LNMO). By capturing a series of time-resolved chemical maps, a new kinetic mechanism for the LNMO phase transformation has been determined [3, 4].

References:

[1] Lu, P. *et al*, *Sci. Rep.* **4**, 3945 (2014).

[2] Lu, P. *et al*, *Microsc. Microanal.* **20**, 1782-1790 (2014).

[3] Lu, P. *et al*, submitted to *Nano Letters*.

[4] Sandia National Laboratories is a multi-program laboratory managed and operated by Sandia Corporation, a wholly owned subsidiary of Lockheed Martin Corporation, for the US Department of Energy's National Nuclear Security Administration under contract DE-AC04-94AL85000.

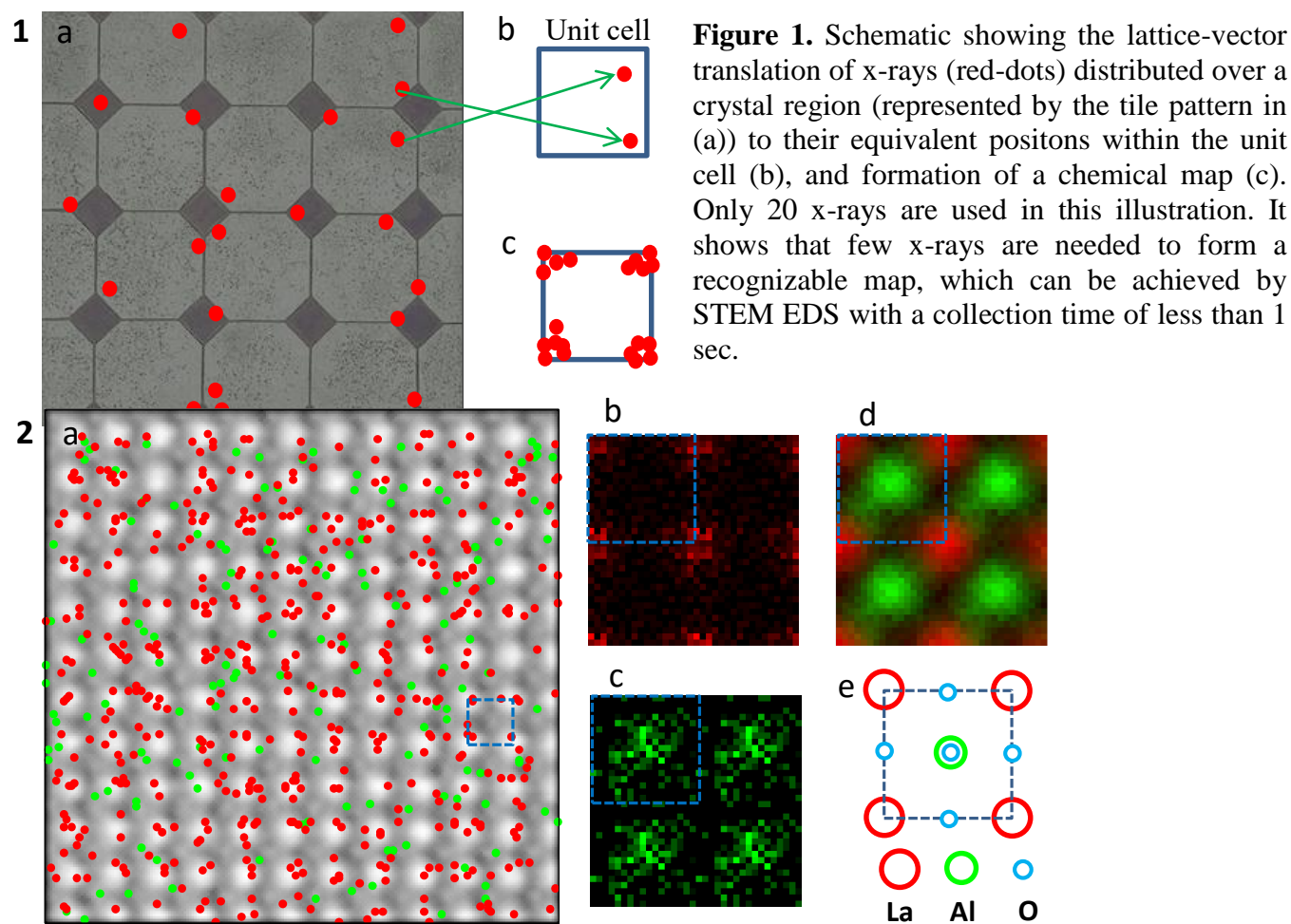


Figure 2. (a) raw x-ray count maps of La L (red-dot) and Al K (green-dot), obtained experimentally by STEM EDS with a total acquisition of 1.8 sec, overlaid on the reference HAADF image acquired simultaneously; (b, c) x-ray count maps (2×2 unit-cells shown) for La L, and Al K x-rays, obtained by the lattice-vector translation method using the x-ray maps in (a); (d) (La, Al) RGB map obtained with the maps in (b) and (c) after 3-point smoothing; and (e) projected LAO structure in [001] direction. A FEI TitanTM G2 80-200 STEM with a Cs probe corrector and ChemiSTEMTM technology operated at 200 kV was used in this study.