

## Microstructural Evolution of High-Strain-Rate Severe Plastic Deformation Processed 316L during Kr Ion Irradiation and elevated Temperature Exposures

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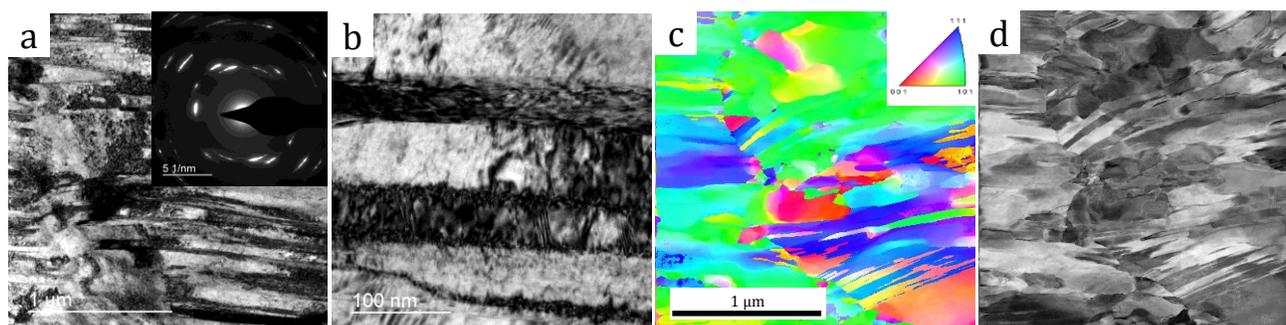
Austenitic stainless steels (SS) are used as structural materials in reactor internal components of nuclear power plants and also in fast reactors [1]. They degrade by prolonged elevated temperature exposure to neutron irradiation, leading to radiation enhanced kinetics for point defect migrations, manifest in hardening, embrittlement and void swelling, irradiation creep, radiation-induced segregation (RIS) and precipitation [1, 2]. RIS accelerated thermally-driven phase transformations are expected to become increasingly relevant for the degradation of austenitic SS under extended life conditions and development of materials with improved irradiation-damage tolerance is desirable [2]. Enhanced irradiation damage tolerance has been reported for materials with microstructures comprising high densities of sink sites for annihilation of excess point defects, e.g. *intra-granularly* provided by thermodynamically stable precipitates and *inter-granularly* as grain boundaries [3-7].

Here we report a transmission electron microscopy (TEM) based investigation of the microstructural development of 316L SS alloys modified by novel high-strain rate severe plastic deformation (SPD) methods based on linear plain strain processing under exposures to Kr ion-irradiation and elevated temperatures [8-11]. The SPD-modified SS offer significantly improved mechanical strength, fatigue performance, resist thermal coarsening up to  $\sim 650^{\circ}\text{C}$ , and exhibit highly deformed, ultra-fine-grain to nano-scale refined grains of austenitic-Fe with lath-like morphology and trace amounts of martensite [8-11] (Fig. 1). In-situ irradiation TEM experiments were performed with 1MeV Kr ions at 298K, 573K, 673K and 773K for fluence up to  $\sim 1 \times 10^{17}$  ions/cm<sup>2</sup>, equivalent to about 20dpa, using the IVEM-Tandem Facility at Argonne National Laboratory. In-situ sequences during heating and ion-irradiation have been recorded typically with a frame rate of 10 frames per second with 300keV electrons using a double-tilt heating holder. TEM analyses prior to and after the ion-irradiation exposures have been performed using a Jeol JEM2100F and a FEI Tecnai G2 F20ST operated at 200kV. Experimental details for automated crystal orientation mapping (ACOM), which has been performed with NanoMegas ASTAR and TOPSPIN systems, and for TEM specimen preparation have been reported previously [8-11]. The 316L SS microstructure remained austenitic, resisted significant grain coarsening, retained its morphology and did not exhibit any precipitation of intermetallic or carbide phases after up to  $\sim 6.5$ h of Kr ion irradiation ( $\sim 20$ dpa) at up to 773K (e.g. Fig. 2). In-situ TEM revealed the formation of interstitial dislocation loops and interactions of the irradiation induced defects with each other and the pre-existing intra-granular dislocation structures and the grain boundaries, leading to reduction in dislocation density and local grain disorientations. The dynamic in-situ TEM observations enabled identification of different fluence dependent regimes during irradiation damage evolution for the different temperatures ranging from 298K to 773K. Complementing TEM based analyses prior to and after ion irradiation enabled quantitative measurements of grain size, locally nano-meter scale resolved grain orientations and grain boundary network structure. The microstructure evolution of the modified 316L during irradiation has been

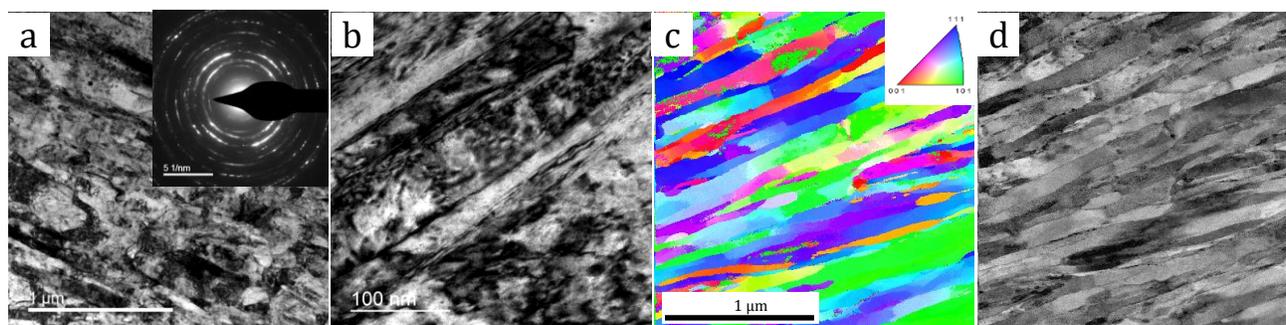
studied with a specific focus on the role of changes to the grain boundary structure and the intragranular dislocation populations. [12]

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**Figure 1.** Bright field TEM (a), (b), precession electron diffraction based (ACOM) orientation map (c), and indexing reliability map (d), for the modified 316L prior to irradiations.



**Figure 2.** Bright field TEM (a), (b), precession electron diffraction based (ACOM) orientation map (c), and indexing reliability map (d), for the modified 316L after 573K Kr-ion irradiations to ~20dpa.