Isotropic to Anisotropic Transition Observed in Si Nanoparticles Lithiation by in situ TEM

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Si is a promising candidate anode material for the next generation Li-ion battery because of its high capacity. However, Si inherently suffers from its large volume change (~300%) when alloyed with Li during the typical charge/discharge cycles, which degrades battery performance and causes potential safety issues. Understanding of microstructural evolutions of Si in the lithiation/delithiation cycles is helpful to design better electrodes. In-situ TEM has been recently applied to study dynamically microstructural evolution of pure Si in the electrochemical lithiation/delithiation reaction [1-6]. Crystalline Si undergoes a two-phase process upon the first lithiation, with an interface between amorphous Li_xSi phase that is formed during the alloying reaction of Si and Li, and the remainder (unreacted) crystalline Si in the center. The lithiation of amorphous Si has similar two-phase process, where the interface is defined between Li-poor core and Li-rich shell regions. The first lithiation of crystalline Si is anisotropic and dominated by the alloying reaction at the interface [2-4]. The first lithiation of amorphous Si nanoparticle is, however, isotropic [1]. The critical threshold of generating fracture for amorphous Si nanoparticles (870 nm) is much higher than that of the crystalline ones (150 nm) [1].

We studied the lithiation of crystalline Si nanoparticles encapsulated by the graphene sheets (Si/Graphene) by in-situ TEM [5]. Unexpectedly, in the first lithiation process, crystalline Si nanoparticles undergo an isotropic to anisotropic transition, which is not observed in pure crystalline and amorphous Si nanoparticles. As shown in fig. 1a-f, the reaction interface between the core (crystalline Si) and shell (amorphous Li_xSi) changes from isotropic to anisotropic as the lithiation proceeds. Such a surprising phenomenon arises from the uniformly distributed localized potential around the Si nanoparticles due to the highly conductive graphene sheets, as schematically shown in fig.1g. The phase transformation of Si in the *in-situ* lithiation/delithiation reactions has also been studied by electron diffraction. Upon first lithiation, a core-shell structure with crystalline Si core and amorphous Li_xSi shell is formed. As shown in Fig.2a, the diffraction pattern taken from a partially lithiated Si/Graphene nanoparticles consists of diffraction spots from crystalline Si (core) and diffuse diffraction rings from amorphous Li_xSi shell. At the late stage of the first lithiation, a phase transformation from amorphous Li_xSi to nano-sized crystal Li₁₅Si₄ can be identified as shown in fig.2b-c.

Unlike pure crystalline Si nanoparticle, the lithiation of Si/Graphene nanoparticle is isotropic at the beginning. This could lead to a more uniformly distributed stress in the nanoparticle thus improvement in cycling stability, similar to that with amorphous Si nanoparticles. These findings offer guidance to designing of improved nanostructured Si materials as fast and large-capacity anode materials for lithium ion batteries.

References:

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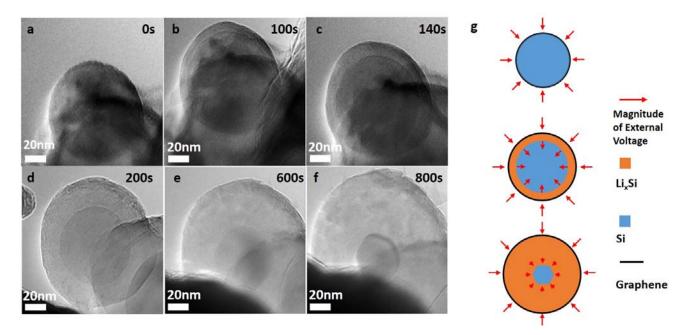


Figure 1. The first lithiation of a single graphene sheet encapsulated Si nanoparticles with lithiation time of (a) 0s, (b) 100s, (c) 140s, (d) 200s, (e) 600s and (f) 800s; (g) schematic of isotropic to anisotropic lithiation of crystalline Si nanoparticle. The magnitude of localized external voltage/work applied at the Si/Li_xSi interface (reaction front) decreases along with the lithiation as indicated by read arrows.

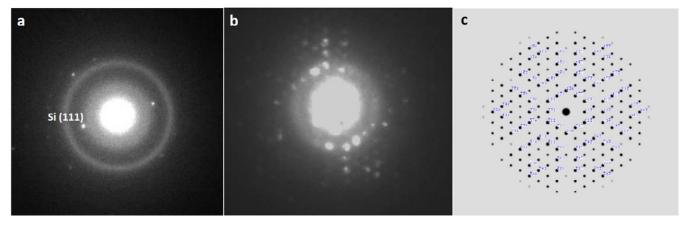


Figure 2. Phase transformation in the lithiation process. (a) Selected-area electron diffraction of a partial lithiated Si/graphene nanoparticles, showing amorphous ring of Li_xSi and diffraction spots of the Si core. (b) A nano-beam electron diffraction collected from a area in the fully lithiated Si/Graphene nanoparticle, which can be indexed as [111] zone axis of Li₁₅Si₄ crystal. (c) Simulated SAED pattern of Li₁₅Si₄ crystal along the [111] axis.